# In Situ Real-Time Observation of Photoinduced Nanoscale Azo-Polymer Motions Using High-Speed Atomic Force Microscopy Combined with an Inverted Optical Microscope

Keishi Yang, Feng-Yueh Chan, Hiroki Watanabe, Shingo Yoshioka, Yasushi Inouye, Takayuki Uchihashi, Hidekazu Ishitobi, Prabhat Verma, and Takayuki Umakoshi\*



under light irradiation. Here, we demonstrate the in situ observation of nanoscale azo-polymer motion by combining tipscan HS-AFM with an optical system, allowing HS-AFM observations precisely aligned with a focused laser position. We



observed the dynamic evolution of unique morphologies in azo-polymer films. Moreover, real-time topographic line profile analyses facilitated precise investigations of the morphological changes. This important demonstration would pave the way for the application of HS-AFM in a wide range of research fields.

**KEYWORDS:** high-speed atomic force microscopy, azo-polymer, photoresponsive material, nanoscale motion, in situ measurement, high spatiotemporal resolution

j igh-speed atomic force microscopy (HS-AFM) is a welligh-speed atomic force interester, recognized tool for capturing biological dynamic motions, including conformational changes in proteins with high spatiotemporal resolutions,<sup>1</sup> enabled by recent technical developments such as small cantilevers, fast scanners, and dynamic feedback control. It has contributed significantly to numerous biological discoveries by providing movies of molecular dynamics at the single-molecule level, including walking myosin V,<sup>2</sup> rotary motion of rotorless  $F_1$ -ATPs,<sup>3</sup> DNA cleavage by CRISPR-Cas9,<sup>4</sup> etc.<sup>5-11</sup> It has already become an indispensable technique in the field of biology.

In addition, recent technical developments in HS-AFM have even improved its analytical capability, which has expanded the possibilities for its widespread use in various fields.<sup>12-15</sup> Among several technical improvements, tip-scan-type HS-AFM is greatly important as a critical development.<sup>16,17</sup> In the original HS-AFM, the tip is fixed while the sample stage is scanned rapidly. In contrast, tip-scan HS-AFM is a stand-alone system for scanning the tip, which can be mounted on an inverted optical microscope.<sup>17</sup> It also enabled the implementation of complex sample manipulation, such as uniaxial stretching.<sup>15</sup> It was originally aimed at biological applications for studying complex phenomena by combining HS-AFM with optical microscopes.<sup>16–18</sup> However, the superior capability of tip-scan HS-AFM can contribute to various other fields such as material science and physical chemistry. It holds great promise as a technique for providing powerful and unique measurements via its incorporation with various optical techniques. For example, by irradiation of a sample with a tightly focused laser and precise adjustment of the tip position to the focused laser spot, the dynamic processes locally induced in novel photoactive materials can be observed at the nanoscale in real time.

As novel photoactive materials, azo-polymers are greatly important in material science. Azo-polymers are photoresponsive organic polymers that include functional azobenzene moieties as side chains of their backbone chains. It induces photoisomerization reactions between the trans and cis forms under light irradiation, as shown in the inset of Figure 1. When

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Figure 1. Schematic of the experimental setup for the in situ HS-AFM observation of azo-polymer deformation.

made into thin films, they exhibit a unique morphology on their surface because of the polymer movements caused by photoisomerization reactions.<sup>19-26</sup> Therefore, they have been considered to be promising materials for various optical applications such as optical data storage, grating-based devices, and photomechanical actuators.<sup>27-29</sup> However, the detailed mechanisms of the deformation process of azo-polymer films, which are crucial for fully utilizing the unique properties of azopolymers for actual applications, remain elusive. In previous studies, conventional AFM has been used to study the azo-polymer deformation process.<sup>21–26</sup> To understand the temporal changes in the surface morphologies, distinct deformed surfaces were created at varying laser exposure durations. After laser irradiation, AFM images of these pre-deformed structures were captured as time-lapse sequences to indirectly investigate temporal evolution. However, this indirect method provides limited information. Unfortunately, it is also technically difficult to analyze a process with a high temporal resolution. The *in situ* observation of azo-polymer movement would provide more reliable insights to elucidate the deformation mechanism.

In this study, we applied tip-scan HS-AFM to record a realtime movie of the photoinduced nanoscale movements of an azo-polymer film. Compared with conventional AFM, HS-AFM enables direct in situ observation of azo-polymer deformation with high temporal resolution, which allows visualization of the formation of individual structures in real time. To this end, we constructed a tip-scan HS-AFM instrument combined with an optical setup. In the optical setup, the laser focus position could be precisely adjusted to the tip position to perform HS-AFM imaging, while the photoactivated azo-polymer motions were induced by a tightly focused laser. We successfully demonstrated in situ HS-AFM observations of azo-polymer movements with high spatiotemporal resolution. We also confirmed that the azopolymers formed different surface morphologies, depending on the direction of polarization of the incident light. Furthermore, we found that the morphologies were slightly different from each other, even when they were created under the same experimental conditions, highlighting the importance of observing individual structures using HS-AFM, unlike the assembly of AFM images of several different structures. These results indicate that tip-scan HS-AFM is highly beneficial for elucidating the mechanism of azo-polymer deformation. We also believe that tip-scan HS-AFM can be a significant tool for researchers to utilize in their own research fields, from material science to many other fields.

To observe azo-polymer movements, a tip-scan HS-AFM instrument was custom-built (Figure 1), on the basis of a previous study.<sup>19</sup> A miniaturized microcantilever (Olympus, BL-AC10DS-A2), 9  $\mu$ m in length, 2  $\mu$ m in width, and 100 nm in thickness, was used for high-speed imaging. The resonance frequency of this cantilever in air is ~1500 kHz. The stand-alone tip-scan HS-AFM instrument was mounted on an inverted optical microscope (Nikon, ECLIPSE Ti2). The incident laser beam (Cobolt, 0532-04-01-0100-700,  $\lambda = 532$  nm), expanded using a beam expander, was focused from the bottom side of the samples through an oil-immersion objective lens (Olympus, NA  $1.45, 100 \times$ ) of the inverted optical microscope. The opposite surfaces of the samples were probed by HS-AFM with the tip set at the top, as shown in the inset of Figure 1. The power of the incident light was adjusted to  $\sim$ 4 nW at the sample plane by using ND filters. Such a low laser power does not cause any photodamage to the samples. The focal size was ~220 nm due to the diffraction limit. Waveplates and polarizers were used to control the polarization of the incident light. To observe the surface deformation process using HS-AFM, the position of the focus spot was precisely adjusted to the position of the tip. For this, homemade piezo-mirror scanners were installed to precisely control the position of the focal spot in the lateral direction. Another piezo scanner was installed on the sample stage to change the position of the sample. A mechanical shutter (Thorlabs, SHB05) was used to control the laser exposure time. All HS-AFM measurements were conducted in air.

An azo-polymer film was prepared from poly(4'-{[2-(methacryloxy)ethyl]ethyl}amino-4-nitroazobenzene) (PMA-DR1). PMA-DR1 (Sigma-Aldrich, 579009) was dissolved in chloroform, and the mixture stirred for 24 h at room



**Figure 2.** (a) HS-AFM images of an azo-polymer film under laser irradiation, obtained at 2 fps. (b) HS-AFM images of an azo-polymer film under laser irradiation with the polarization direction rotated by  $90^{\circ}$  with respect to that in the case of panel a. Green arrows indicate the direction of polarization of the incident laser.



**Figure 3.** (a) HS-AFM image of the azo-polymer film, which is the same as that shown in Figure 2a, for an irradiation time of 14 s. (b) Height line profiles obtained along the white arrow in panel a, obtained for all frames of the HS-AFM movie at the same line position to investigate the temporal change. (c) Changes in the temporal height of the two lobes and pit, analyzed from panel b. (d) Temporal change in the distance between the two lobes, analyzed from panel b. (e) HS-AFM image that is the same as that shown in Figure 2b for an irradiation time of 14 s. The incident polarization was rotated by 90° with respect to that in panel a, as indicated by the green arrows. (f) Height line profiles obtained along the white arrow in panel e, obtained for all frames of the HS-AFM movie. (g) Changes in the temporal height of the two lobes and the pit, analyzed from panel f. (h) Temporal change in the distance between the two lobes, analyzed from panel f.

temperature. The PMA-DR1 solution was filtered twice through a membrane filter with a pore size of 200 nm. The solution was spin-coated on a coverslip at 300 rpm for 3 s and subsequently at 1500 rpm for 60 s. The spin-coated film was heated in an oven for 1 h at 110 °C to remove the remaining solvent. The thickness of the azo-polymer film was ~30 nm.

Using the constructed setup, azo-polymer deformation was successfully observed in real time at an imaging rate of 2 frames per second (fps), as shown in Figure 2a (Movie S1). Linearly polarized light was used in this experiment; the polarization direction is indicated by the green arrow in Figure 2a. The focus position and size are indicated by the green circle. Upon initiation of light irradiation at 0 s by opening the mechanical shutter, the sample surface, which was initially flat, began to display two lobes aligned with the polarization direction of the incident light. A pit also formed between the two lobes. This unique pattern is similar to that reported in a previous study,<sup>22</sup> in which conventional AFM was used after a complete exposure. This confirmed that in our measurements, where we continuously measure AFM images during the exposure, azopolymer movements can be observed with high temporal resolution using the HS-AFM technique. Over time, the two lobes moved higher, and the pit became deeper, with the left

lobe being slightly higher than the right lobe. In addition, the distance between the lobes increased. This change in the surface morphology during the exposure, which is observed for the first time, is attributed to the azo-polymer moving outward from the center of the focus spot along the polarization direction of the incident light, as previously reported.<sup>22</sup> With linearly polarized light irradiation, the azobenzene molecules are oriented in the direction perpendicular to the incident light polarization during *trans*  $\leftrightarrow$  *cis* photoisomerization.<sup>30</sup> According to the anisotropic nature induced by the molecular orientation, anisotropic photofluidic force is generated under the condition that the polymer is softened by the repeated photoisomerization reactions between the *trans* and *cis* forms.<sup>31</sup> This anisotropic photofluidic force induces the formation of an anisotropic pattern depending upon the incident polarization;<sup>32</sup> however, a more detailed mechanism is still under investigation. Next, we observed azo-polymer deformation by rotating the polarization direction by 90°, as indicated by the green arrow in Figure 2b (see also Movie S2). As expected, the azo-polymer film formed a structure similar to that shown in Figure 2a, but the pattern was rotated by  $90^{\circ}$ . In this case, the heights of the two lobes were almost the same with better symmetry in the pattern.

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**Figure 4.** (a) HS-AFM images of the azo-polymer film under laser irradiation with the polarization perpendicular to the film (*z*-polarization). (b) Height line profile along the white arrow shown in panel a for all frames of the HS-AFM movie. (c) Changes in the temporal height of the lobe and pit.

The topographic line profiles based on the HS-AFM movies allow a detailed and quantitative analysis of the temporal changes in deformation with high spatiotemporal resolution. Figure 3a shows the HS-AFM image with an irradiation time of 14 s, which is identical with that of Figure 2a. The overlaid height line profiles from all frames along the white arrow in the image are shown in Figure 3b (see also Movie S3). The line profile analysis quantitatively visualized the changes in the surface topography, showing the gradual formation of two lobes and a pit between them immediately after laser irradiation, where their motions are indicated by the arrows. The left lobe was twice as high as the right lobe. To investigate the changes in more detail, the heights of the two lobes and pit were plotted as a function of the laser irradiation duration, as shown in Figure 3c. The heights of the lobes increase linearly with estimated growth rates of 1.60 nm/s for the left lobe and 0.91 nm/s for the right lobe. They eventually increased to 24.0 nm for the left lobe and 14.1 nm for the right lobe after laser irradiation for 14 s. The height at the pit position was almost the same or slightly increased in the first few frames, as also seen in the line profiles in Figure 3b. Subsequently, the height gradually and linearly decreased to approximately -4.5 nm after the laser exposure for 14 s. We assume that this is because the two lobes were close to each other at the beginning, resulting in an increase in the height at the center. As the two lobes moved away from each other, the height of the center part decreased, leading to the formation of the pit. Figure 3d shows the temporal changes in the distance between the two lobes. Notably, this progression exhibited a nonlinear change; the lobes moved away rapidly in the beginning and decelerated subsequently. Given the diffractionlimited focal spot size of  $\sim$ 220 nm, the lobes moved faster when they were initially close to the focal spot. They slowed as they moved farther from the focal spot. The final distance between the lobes was 395 nm after 14 s. We conducted the same analysis for the HS-AFM images shown in Figure 2b, in which the polarization direction was rotated by 90°. Figure 3e shows the HS-AFM image of the azo-polymer film, shown in Figure 2b for an irradiation time of 14 s. The height line profiles along the white arrow in Figure 3e are shown in Figure 3f (see also Movie S4). In this case, the height line profiles of the two lobes were symmetrical. The upper and lower lobes grew at rates of 1.02 and 0.97 nm/s, respectively, as shown in Figure 3g. Their heights eventually increased to 14.3 nm for the upper lobe and 14.1 nm for the lower lobe. With regard to the change in height at the position of the pit, we observed behavior similar to that shown in Figure 3c. As shown in Figure 3h, a similar nonlinear change was also observed in the distance between the lobes, although the

final distance was 325 nm, which is slightly different from the case shown in Figure 3d. Thus, HS-AFM can accurately and quantitatively analyze how individual azo-polymer structures are formed in real time, which is difficult to achieve using conventional AFM. These results clearly demonstrate the usefulness of HS-AFM observations. It should also be emphasized that the two structures shown in panels a and e of Figure 3 were created under the same experimental conditions, except for the polarization direction; however, the resulting structures were slightly different. For example, the pattern in Figure 3e was symmetric whereas that in Figure 3a was asymmetric. The distance between the two lobes also differed between the two cases. We assume that this difference is due to a slight difference in the focal position or optical alignment. Although we attempted to maintain the same experimental conditions as much as possible, even a very small difference in the beam profiles of the laser focus can drastically affect the deformation pattern.<sup>23</sup> Because a small difference can cause different structures, it is highly beneficial to observe the deformation process of individual structures using HS-AFM, rather than obtaining the averaged information about several different structures by conventional AFM.

In addition to the linear polarization parallel to the substrate, we observed the deformation process with *z*-polarization, which is a linear polarization perpendicular to the substrate. We used a *z*-polarizer (ZPol, Nanophoton) to generate the *z*-polarization. Figure 4a shows the clipped HS-AFM images demonstrating the change in the surface morphology of the azo-polymer film under laser irradiation with *z*-polarization (see also Movie S5). Unlike horizontally polarized laser irradiation, a single lobe appeared at the center and a concentric circular pit surrounding the lobe was formed, which is consistent with a previous study.<sup>25</sup> Over time, the center lobe became higher and the concentric circular pit became deeper. As shown in Figure 4b (see also Movie S6), we quantitatively analyzed the deformation dynamics using the height line profile obtained along the white arrow in the image in Figure 4a. Here, pit1 and pit2 indicate the bottom and top parts, respectively, of the concentric pit, as indicated in Figure 4a. The time evolution of the lobe height is plotted in Figure 4c, showing a monotonic and linear increase at a rate of 0.63 nm/s. It eventually increased to 8.7 nm. The pit height decreased monotonically and linearly at a rate of -0.29 nm/s, finally reaching -4.0 nm after 14 s in the pit1 and pit2 regions. In addition, we conducted the same experiment several times to investigate the variations in the changes in the temporal height between the different lobes (Figure S1). We found that there were differences of a few nanometers between the heights of

each lobe, even though they were created under identical conditions, which again highlights the importance of analyzing the individual structures by using HS-AFM.

We succeeded in observing the deformation of the azopolymer film by HS-AFM. However, even though HS-AFM has been recognized as a gentle technique, owing to the soft cantilever and sophisticated control system such as dynamic proportional-integral-differential (PID) feedback, such that even fragile proteins can be observed with a negligible influence on their functions, high-speed scanning of the AFM tip may affect the deformation of the azo-polymer and the resulting structure. Thus, we evaluated whether tip scanning affected the deformation of the azo-polymer film. We induced azo-polymer deformation for 15 s using a z-polarized incident laser while observing the process by HS-AFM. In contrast, as a control experiment, deformation was induced for 15 s without HS-AFM observation. HS-AFM observation was performed after laser irradiation for 15 s, which means that high-speed tip scanning did not affect deformation. The same experiment was repeated three times for each condition. Panels a and c of Figure 5 show



**Figure 5.** (a) HS-AFM images of the azo-polymer film after deformation, in which HS-AFM observations were conducted during deformation. (b) Height line profiles obtained along the lines marked in panel a. (c) HS-AFM images of the azo-polymer film after deformation in which HS-AFM observations were conducted after deformation. (d) Height line profiles obtained along the lines in panel c.

HS-AFM images of the azo-polymer after deformation, with and without HS-AFM observations, respectively, during the deformation process. Although all HS-AFM images showed slight differences because a small difference in the experimental conditions could affect the resulting structure, the overall surface morphologies were considerably similar. Panels b and d of Figure 5 show the height line profiles along the arrows marked on each HS-AFM image. Although random differences were observed among the line profiles, they were considerably similar. The average height of the lobes was  $\sim 10$  nm in both cases regardless of whether HS-AFM observations were conducted during deformation. Therefore, we experimentally verified that HS-AFM can be applied to the observation of photoinduced movements of organic molecules such as azo-polymers.

In this study, we successfully demonstrated the *in situ* realtime observation of the photoinduced surface deformation of an azo-polymer thin film using tip-scan HS-AFM combined with an inverted optical microscope. The HS-AFM movie visualized the photoinduced changes in the surface morphology with high spatiotemporal resolution, showing the formation of unique patterns depending on the direction of polarization. Moreover, the time evolution of the height line profile enabled precise and quantitative analysis. We found that the height of the center area increased in the first moment before it decreased to form a pit in the case of linear polarizations. In addition, we found that a possible slight difference in the experimental conditions, such as optical alignment, significantly affected the resulting morphology. This fact suggests the importance of continuous observation of individual structures from the beginning to the end of deformation by HS-AFM, which is challenging with conventional AFM. We believe that these findings revealed by *in situ* HS-AFM observations will contribute to further elucidating the mechanisms of azo-polymer deformation.

Although we succeeded in observing the dynamic process of azo-polymer deformation, the imaging rate was still limited to 2 fps in this study. This is because the HS-AFM measurements had to be performed in air as the current sample was not very resistant to water. In general, the quality factor of the cantilever in air is several hundreds or tens of times higher than that in solution, which slows the transient response of the cantilever and thus hinders high-speed imaging. By observing deformation in water, we would be able to capture faster dynamic processes in detail under different conditions, such as a high laser power causing rapid movements. More details of the deformation mechanism can be further understood through HS-AFM observations under various conditions, such as different laser wavelengths, polarization conditions, and types of azo-polymer molecules. For example, previous reports have shown the formation of unique spiral relief on azo-polymer films using an optical vortex.<sup>24–26</sup> Furthermore, this technique would be useful for not only azo-polymers but also many other samples in different fields. The potential of tip-scan HS-AFM has been extended to various fields by combining it with optical techniques. Our important demonstration of tip-scan HS-AFM would significantly stimulate researchers in diverse research fields, contributing to novel discoveries in the future in research fields from biology to many others, including material science and physical chemistry.

## ASSOCIATED CONTENT

#### Supporting Information

The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/acs.nanolett.3c04877.

- Legends of HS-AFM movies of azo-polymer deformation and changes in the temporal height of lobe structures (PDF)
- HS-AFM movie of the azo-polymer deformation process with linearly polarized light shown in Figure 2a (Movie S1) (MP4)
- HS-AFM movie of the azo-polymer deformation process with linearly polarized light shown in Figure 2b (Movie S2) (MP4)
- Time evolution of the height line profiles shown in Figure 3b (Movie S3) (MP4)
- Time evolution of the height line profiles shown in Figure 3f (Movie S4) (MP4)
- HS-AFM movie of the azo-polymer deformation process with *z*-polarized light (Movie S5) (MP4)
- Time evolution of the height line profiles shown in Figure 4b (Movie S6) (MP4)

### AUTHOR INFORMATION

#### **Corresponding Author**

Takayuki Umakoshi – Department of Applied Physics, Osaka University, , Suita, Osaka 565-0871, Japan; Institute for Advanced Co-Creation Studies, Osaka University, Suita, *Osaka 565-0871, Japan;* orcid.org/0000-0002-0479-1124; Email: umakoshi@ap.eng.osaka-u.ac.jp

## Authors

- Keishi Yang Department of Applied Physics, Osaka University, , Suita, Osaka 565-0871, Japan
- Feng-Yueh Chan Department of Physics, Nagoya University, Nagoya, Aichi 464-8602, Japan
- Hiroki Watanabe Exploratory Research Center on Life and Living Systems (ExCELLS), National Institutes of Natural Sciences, Okazaki 444-8787, Japan
- Shingo Yoshioka Department of Applied Physics, Osaka University, , Suita, Osaka 565-0871, Japan
- Yasushi Inouye Department of Applied Physics, Osaka University, , Suita, Osaka 565-0871, Japan; Graduate School of Frontier Biosciences, Osaka University, Suita, Osaka 565-0871, Japan; Occid.org/0000-0001-5118-5146
- Takayuki Uchihashi Department of Physics, Nagoya University, Nagoya, Aichi 464-8602, Japan; Exploratory Research Center on Life and Living Systems (ExCELLS), National Institutes of Natural Sciences, Okazaki 444-8787, Japan; ◎ orcid.org/0000-0002-0263-5312
- Hidekazu Ishitobi Department of Applied Physics, Osaka University, , Suita, Osaka 565-0871, Japan; Graduate School of Frontier Biosciences, Osaka University, Suita, Osaka 565-0871, Japan; Occid.org/0000-0002-1145-6850
- Prabhat Verma Department of Applied Physics, Osaka University, , Suita, Osaka 565-0871, Japan; Ocid.org/ 0000-0002-7781-418X

Complete contact information is available at: https://pubs.acs.org/10.1021/acs.nanolett.3c04877

#### **Author Contributions**

T. Umakoshi conceived and designed this project. K.Y. performed all experiments, analyzed the results, and wrote the manuscript. K.Y., F.-Y.C., H.W., S.Y., T. Uchihashi, and T. Umakoshi constructed the setup for experiments. H.I. prepared the sample. H.I. and T. Umakoshi supervised this research. All authors contributed to the discussion and finalization of the manuscript.

#### Notes

The authors declare no competing financial interest.

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